



Introduction

Since its theoretical prediction in the 1940s and experimental isolation from graphite in 2004, graphene has rapidly emerged as a highly desirable quantum material in various application areas, including semiconductors, bioengineering, microelectronics, batteries, and composites.¹² Producing large-scale, defect-free, layer-uniform graphene devices is crucial for many applications. Pristine monolayer graphene exhibits excellent mechanical, thermal, and electronic properties that change with increasing layer numbers, strain, and defects.³ Methods to probe uniformity across graphene devices are therefore crucial for quality assurance.

The Edinburgh Instruments RM5 Confocal Raman Microscope is ideal for analysing graphene. Raman mapping reveals information on graphene's structural and electronic properties. It can map the location of structural defects, strain, doping, and layer stacking using the relative intensities and positions of the Raman bands in the graphene spectrum. In this Application Note, the capability of the RM5 and Ramacle[®] software for graphene analysis is demonstrated by mapping the layer number, strain, and defects on graphene films.

Materials and Methods

The samples analysed were monolayer and bilayer graphene films deposited on Si substrates. Raman mapping was performed using an Edinburgh Instruments RM5 equipped with a 532 nm laser, a 100X 0.9 NA objective, and a CCD detector, Figure 1. All mapping data was analysed and plotted using the RM5's Ramacle software package.



Figure 1. Edinburgh Instruments RM5 Raman Microscope.

Raman Spectrum of Graphene

The Raman spectrum of graphene with the key bands for mapping is shown in Figure 2. The D band *ca*. 1350 cm⁻¹ is a ring breathing mode of the sp² carbon ring and is Raman active in defective graphene. The G band *ca*. 1585 cm⁻¹ is caused by an in-plane stretching vibrational mode of the sp² bonds, and the 2D band *ca*. 2680 cm⁻¹ originates from a double resonance enhanced two-phonon lateral vibrational process.⁴



Figure 2. Raman spectrum of graphene with the common modes used for mapping labelled.



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Layer Number: Ratiometric Mapping of G and 2D Bands

The variation in layer number across a graphene sample can be determined from the relative peak intensities of the G band (I_G) and the 2D band (I_{2D}) in the Raman spectrum.⁵ As the number of layers in graphene increases, I_G increases relative to I_{2D} .⁶ The 2D/G band ratio (I_{2D}/I_G), therefore, can map the uniformity and layer number across graphene flakes and films.

A nominally bilayer graphene sample was mapped using the RM5. Maps of I_G , I_{2D} , and the peak ratio I_{2D}/I_G were plotted in Ramacle and are shown in Figure 3. The I_{2D}/I_G peak ratio varies significantly across the map between 0.1 and 2 (Figure 3c), indicating a significant layer number variation across the

graphene sample. Monolayer graphene is typically characterised by an I_{2D}/I_G of ≥ 2 , while in bilayer graphene, I_{2D}/I_G approaches 1. Additional graphene layers cause I_{2D}/I_G values to be lower than $1.^{7,8}$. Therefore, the black areas (high I_{2D}/I_G) on the map were assigned as monolayer graphene, the purple layers as bilayer, and the areas in yellow (low I_{2D}/I_G) were assigned as multilayer graphene. The spectra in Figure 3d show the significant change in I_{2D}/I_G in the spectra collected within the different regions.



Figure 3. Izo/IG Raman mapping of graphene for layer number determination.



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Strain: Peak Position Mapping of G and 2D Bands

Strain in materials causes bond lengths to change, altering their vibrational frequencies and the properties of the material under investigation. Raman spectroscopy is, therefore, highly sensitive to strain. In graphene, strain is associated with frequency changes in the vibrational modes. This is reflected in the Raman shift of the G and 2D bands, meaning the position of these bands can be used to create a strain map of the graphene surface.⁷

A monolayer film grown on a Si wafer containing several fissures was mapped to track strain in graphene using Raman microscopy, Figure 4. The area mapped on the sample covered two fissures in the Si substrate, which were hypothesised to affect the mechanical properties of the graphene film deposited on top and induce strain. A peak position image for the G band was constructed, Figure 4b and this clearly showed a decrease in the band frequency in the regions where the graphene lay atop the fissures. The reduction in the G band frequency can be assigned to tensile stress in the graphene layer. Tensile stress elongates the graphene sheet, which lowers the vibrational mode energies, resulting in a shift to lower wavenumbers. Spectra from strained and unstrained regions are shown in Figure 4c, where it can also be seen that the frequency of the 2D band decreases in the strained graphene.

The frequency decrease of the G band in Figure 4b could also be caused by an increase in graphene layer numbers. However, I_G would increase with increasing layer number, and Figure 4a shows that areas with a G band frequency decrease also have a reduced I_G . Therefore, it can be concluded that the peak shifts are due to strain and not layer number. Interestingly, the intensity map also shows regions of high I_G , shaded in cream, which could be assigned as multilayer graphene.



Figure 4. Peak position Raman mapping of graphene for strain detection.



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Defects: Ratiometric Mapping of D and G Bands

Raman microscopy can track defects on graphene samples because disorder in the lattice creates D and D' Raman peaks ca. 1350 cm⁻¹ and 1640 cm⁻¹, which would otherwise be Ramanforbidden. These are allowed in spectra of defective graphene because the defect provides the missing momentum to satisfy conservation in the Raman scattering process.⁸ The number and nature of defects in graphene can significantly affect the performance, which can either be detrimental or advantageous depending on the intended application of the material. For example, defects have been shown to diminish electron mobility on graphene deposited on substrates, but they have also been reported to enhance the reactivity of graphene-based catalysts.^{9,10}

The graphene sample was mapped on the RM5 and $I_{\rm D}$ and $I_{\rm G}$ were initially plotted in Ramacle, Figure 5a and b (overleaf). $I_D/I_{\rm G}$ was subsequently plotted in Ramacle, and the resulting image is shown in Figure 5c. $I_D/I_{\rm G}$ is a valuable metric for determining pristine and defect-dense graphene regions on a sample. In

pristine graphene, this ratio is typically close to 0, and as the defect density increases, the ratio will increase. In Figure 5c, the dark blue regions were assigned as low-defect-density graphene, and the areas shaded in red and yellow were high-defect-density graphene. Spectra from different regions of the map are shown in Figure 5d.

In the high-defect-density region, the D band is more intense than the G band, whereas in the low-defect-density region, the G band is dominant, and no D band is visible. The D' band is also present in the defect-dense regions and appears as a higher-Raman shift shoulder of the G band. The ratio between the D and D' bands can also be used to discriminate between different defect types; for example, a larger value is associated with sp³-type defects and lower values are associated with vacancy-type defects.⁸



Figure 5. I_D/I_G Raman mapping of graphene for defect detection.

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Conclusion

An Edinburgh Instruments RM5 Confocal Raman Microscope was used to map layer number, strain, and defect density in graphene films. The RM5 is ideal for analysing graphene, revealing a wealth of information about the structural and electronic properties of the graphene layer with high sensitivity and spatial resolution.

References

- 1. P. R. Wallace, *Phys. Rev.*, 1947, **71**, 622-634.
- 2. K. S. Novoselov et al., Science, 2004, **306**, 666-669.
- 3. S. Asad et al., ACS Appl. Mater. Inter., 2022, **5**, 11933-11946.
- 4. P. Venezuela et al., Phys. Rev. B, 2011, 84, 035433.
- 5. X. Yu et al., Sci. Rep., 2018, 8, 4759.
- 6. A. C. Ferrari, Solid State Commun., 2007, 143, 47-57.
- 7. Z. H. Ni et al., ACS Nano, 2008, **2**, 2301-2305.
- 8. A. Eckmann et al., Nano Lett., 2012, 12, 3925-3930.
- 9. Z. H. Ni et al., Nano Lett., 2010, 10, 3868-3872.
- 10. D. W. Boukhvalov et al., Nano Lett., 2008, 8, 4373-4379.



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